

## Melt growth of bulk Bi<sub>2</sub>Te<sub>3</sub> crystals with a natural p–n junction

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Single crystals of Bi<sub>2</sub>Te<sub>3</sub> were grown from Bi–Te melts using the modified Bridgman method. It was shown for the first time that solidification of 61 and 62 mol.% Te melts provides a built-in p–n junction on the cleaved plane of as grown crystals without any post growth treatment. The formation of a p–n junction along the growth crystal was explained by Te segregation. Both p- and n-parts of the ingot have shown high carrier concentrations  $n \approx p \approx 1 \times 10^{19} \text{ cm}^{-3}$  and high carrier mobility  $\sim 10^4 \text{ cm}^2 \text{ V s}^{-1}$  at 4 K. In the transition p–n region, Hall carrier concentration is decreased by two orders of magnitude as a result of intrinsic compensation of carriers.

### Introduction

Bismuth telluride based compounds are considered to be among the best thermoelectric materials operating near room temperature.<sup>1</sup> During the last 5 decades, a large amount of research has focused on the study of the Seebeck coefficient in doped Bi<sub>2</sub>Te<sub>3</sub> in order to increase the performance of thermoelectric devices. Another issue was to check the type of conductivity provided by a given impurity since it is necessary to have separate blocks of n- and p-type elements.

Satterthwaite and Ure Jr.<sup>2</sup> provided the details on the phase diagram of the Bi–Te binary system close to the melting point of Bi<sub>2</sub>Te<sub>3</sub>. The data show that the composition with congruent melting is not a stoichiometric compound but is shifted to the Bi-rich side. That implies that the equilibrium melt for electrically neutral stoichiometric Bi<sub>2</sub>Te<sub>3</sub> should be enriched with tellurium. Fig. 1a shows a part of the phase diagram of the Bi–Te system. The “boundary” melt composition, which separates primary crystallization of p- and n-type regions of Bi<sub>2</sub>Te<sub>3</sub>, was reported to be ~63 mol% Te + 37 mol% Bi.<sup>2</sup> Later, it was shown that polycrystalline buffer material may be used for changing the composition of bulk Bi<sub>2</sub>Te<sub>3</sub> by annealing.<sup>3</sup> Incorporation of foreign atoms into the structure of Bi<sub>2</sub>Te<sub>3</sub> may also result in the inversion of conductivity. For instance, in ref. 4, it was shown that annealing of bulk p-type

Bi<sub>2</sub>Te<sub>3</sub> crystals in selenium vapor transforms the surface layers of the samples to the n-type Bi<sub>2</sub>Te<sub>3-x</sub>Se<sub>x</sub>. In that way, a p–n junction between outer n-layers and the internal Se free part of the sample was formed. The same effect was shown for sulfur which leads to the formation of a Bi<sub>2</sub>Te<sub>3</sub>–Bi<sub>2</sub>Te<sub>3-x</sub>S<sub>x</sub> ( $x \geq 0.12$ ) p–n junction by using heat treatment.<sup>5</sup> Partial substitution of cation by indium also produces the n-type material. This phenomenon was demonstrated for the melt grown Bi<sub>2-x</sub>In<sub>x</sub>Te<sub>3</sub> ( $x \geq 0.1$ ) material in ref. 6.

In recent years, a great deal of attention has been paid to the properties of the tetradymite mineral group showing a novel property of three-dimensional topological insulators.<sup>7,8</sup> Bismuth telluride is among the compounds of this class; theoretically, its surface state consists of a single nondegenerate Dirac cone, while the bulk state remains insulating.<sup>9</sup> However, due to their nonstoichiometry, all as grown Bi<sub>2</sub>Te<sub>3</sub> crystals are of p- or n-type and therefore exhibit metallic behavior in the bulk. And the resulting intrinsic carriers hamper the study and application of the specific surface states.

Sidestepping of this problem may be carried out by using previous data on transformation of conductivity type in Bi<sub>2</sub>Te<sub>3</sub> since the p–n transition should result in the compensation of intrinsic carriers. For instance, the electrical properties can be tuned through a slight increase of Te content in the as grown p-type Bi<sub>2</sub>Te<sub>3</sub> crystal by Te-vapor annealing.<sup>10</sup> Another example is the growth of Bi<sub>2</sub>(Te,Se)<sub>3</sub> mixed crystals. It was shown that Bridgman solidified Bi<sub>2</sub>Te<sub>2</sub>Se<sub>0.995</sub> melt may provide the samples with a positive to negative transition of the Hall coefficient with temperature.<sup>11</sup> The same effects as well as domains with different carrier types were found in the crystal grown from the Bi<sub>2</sub>Te<sub>2</sub>Se melt.<sup>12</sup>

Although the adjustment of tellurium pressure is very promising for the growth of stoichiometric Bi<sub>2</sub>Te<sub>3</sub> thin

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